

Bifunctional Catalysts

**Mn<sub>x</sub>O<sub>y</sub>/NC and Co<sub>x</sub>O<sub>y</sub>/NC Nanoparticles Embedded in a Nitrogen-Doped Carbon Matrix for High-Performance Bifunctional Oxygen Electrodes\*\***

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**Abstract:** Reversible interconversion of water into H<sub>2</sub> and O<sub>2</sub>, and the recombination of H<sub>2</sub> and O<sub>2</sub> to H<sub>2</sub>O thereby harnessing the energy of the reaction provides a completely green cycle for sustainable energy conversion and storage. The realization of this goal is however hampered by the lack of efficient catalysts for water splitting and oxygen reduction. We report exceptionally active bifunctional catalysts for oxygen electrodes comprising Mn<sub>3</sub>O<sub>4</sub> and Co<sub>3</sub>O<sub>4</sub> nanoparticles embedded in nitrogen-doped carbon, obtained by selective pyrolysis and subsequent mild calcination of manganese and cobalt N<sub>4</sub> macrocyclic complexes. Intimate interaction was observed between the metals and nitrogen suggesting residual M–N<sub>x</sub> coordination in the catalysts. The catalysts afford remarkably lower reversible overpotentials in KOH (0.1M) than those for RuO<sub>2</sub>, IrO<sub>2</sub>, Pt, NiO, Mn<sub>3</sub>O<sub>4</sub>, and Co<sub>3</sub>O<sub>4</sub>, thus placing them among the best non-precious-metal catalysts for reversible oxygen electrodes reported to date.

The theoretically high energy density of metal–air batteries makes them very desirable for future energy conversion and storage applications.<sup>[1–4]</sup> However, unlike conventional secondary batteries, their rechargeability still poses many scientific challenges, one of which being the development of efficient catalysts for the oxygen-reduction reaction (ORR)

during the discharge process, and for the oxygen-evolution reaction (OER) during the recharge process.<sup>[3,4]</sup> Finding stable and efficient electrocatalysts which drive both reactions is difficult, because good catalysts for ORR often tend to be poor for OER and vice versa.<sup>[1,4,5]</sup> For example, platinum which is a very good catalyst for ORR is poor for OER, while ruthenium and iridium oxides which are very good for OER are poor for ORR.<sup>[6–8]</sup> Most bifunctional catalysts are prepared by combining good catalysts for ORR and OER to form a composite.<sup>[9]</sup> At present, the premium catalysts of choice for both reactions comprise platinum-group metals,<sup>[6,9]</sup> which is a drawback from the point of view of cost and availability.

Transition-metal oxides which have structural and compositional semblance to the active center of the oxygen evolving complex, such as spinels, show catalytic activity towards both OER and ORR.<sup>[10]</sup> The electrocatalytic potency of most transition-metal oxides is impaired by their poor electronic conductivity, which however, may be improved by doping the oxides with electron donors or by supporting them on conducting materials.<sup>[11]</sup>

Herein, we exploited the unique structural and functional properties of porphyrins and phthalocyanines as precursors to synthesize manganese-based oxide catalysts with exceptionally high dual activity in electrocatalyzing both ORR and OER. The synthesis involved two key steps. In the first step, the macrocyclic complex (MnN<sub>4</sub>) was impregnated with nitrogen-doped carbon (NC) followed by pyrolysis of the MnN<sub>4</sub>/NC composite at 650 °C for 2 h under He. This step was followed by mild calcination of the product in oxygen at 200 °C for 30 min. The resulting catalyst is hereafter denoted as Mn<sub>x</sub>O<sub>y</sub>/NC. During the synthesis, the macrocyclic structures degrade to form among others, spinel Mn<sub>3</sub>O<sub>4</sub> nanoparticles, and Mn–N<sub>x</sub> moieties embedded in a NC matrix. The nitrogen-functionalized carbon groups in NC, particularly, pyridinic, graphitic, and pyrrolic groups act as complementary ORR catalysts in addition to conferring conductivity to the Mn<sub>3</sub>O<sub>4</sub> nanoparticles, thereby furnishing the catalyst the unusual capability to catalyze both ORR and OER with outstanding efficiency.

We observed that catalysts prepared from different porphyrin ligands (Figure 1S in the Supporting Information) show similar structural and electrochemical properties. We therefore used *meso*-tetrakis(4-pyridyl)porphyrinate manganese(III) (Figure 1S) as the precursor for detailed structural and electrochemical investigation of Mn<sub>x</sub>O<sub>y</sub>/NC.

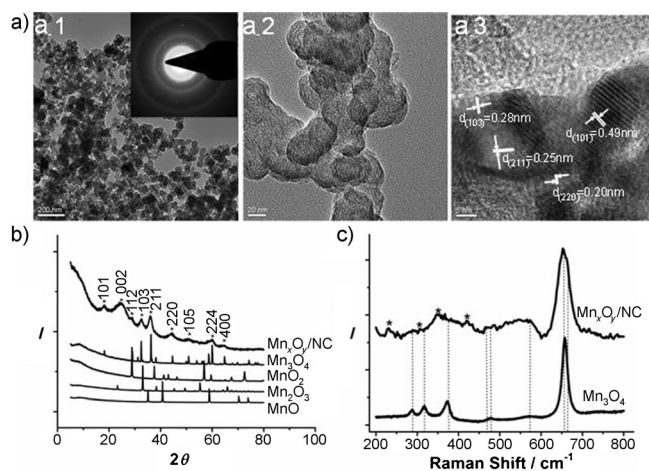
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Representative SEM micrographs of  $\text{Mn}_x\text{O}_y/\text{NC}$  with superimposed energy-dispersive X-ray spectroscopy (EDX) spectra are shown in Figure 2S. The SEM images reveal agglomerated nanoparticles surrounded by an amorphous shell. Figure 1a shows TEM images of  $\text{Mn}_x\text{O}_y/\text{NC}$  at different

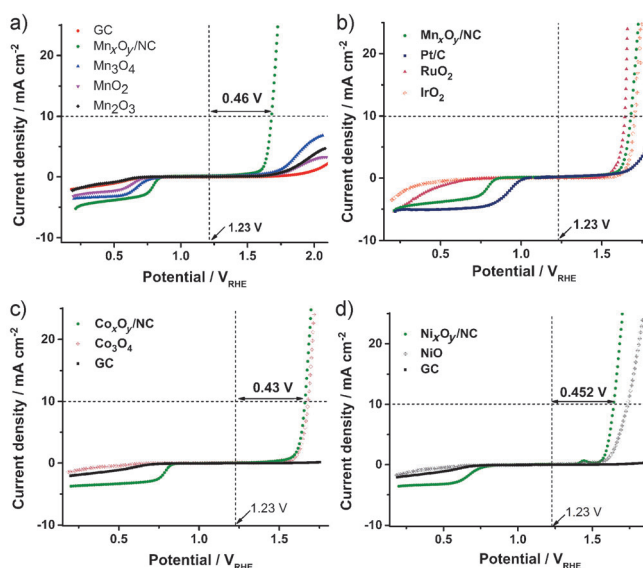


**Figure 1.** a) TEM images of  $\text{Mn}_x\text{O}_y/\text{NC}$  at different magnification scales, b) XRD patterns of  $\text{Mn}_x\text{O}_y/\text{NC}$ , and commercial  $\text{MnO}$ ,  $\text{Mn}_2\text{O}_3$ ,  $\text{Mn}_3\text{O}_4$ , and  $\text{MnO}_2$ , c) Raman spectra of  $\text{Mn}_x\text{O}_y/\text{NC}$  and  $\text{Mn}_3\text{O}_4$  recorded at 661 nm and 10 mW laser power.

magnification scales in which nanoparticles can be seen that are either embedded in or surrounded by amorphous carbon. The selected-area electron diffraction (SAED) image (inset of Figure 1a-1) shows that the particles are polycrystalline. For the embedded particles, the surface of the amorphous carbon shell is highly porous (Figure 1a-2). The diffraction fringes in the high-resolution TEM image of Figure 1a-3 expose the different crystal planes of the particles. The observed lattice spacings of 0.49, 0.20, 0.28, and 0.25 nm correspond to the (101), (220), (103), and (211) crystal planes of tetragonal  $\text{Mn}_3\text{O}_4$ . This result was confirmed by the X-ray diffraction (XRD) results (Figure 1b). The diffraction peak at  $24.57^\circ$  originates from the (002) planes of carbon. The XRD pattern of  $\text{Mn}_x\text{O}_y/\text{NC}$  matches that of  $\text{Mn}_3\text{O}_4$ , space group  $14_1/\text{m}$  and consistent with the JCPDS card file No. 24-0734 thus revealing the formation of crystalline  $\text{Mn}_3\text{O}_4$  particles with a tetragonal hausmannite structure. The Raman spectrum of  $\text{Mn}_x\text{O}_y/\text{NC}$  (Figure 1c) was compared with that of  $\text{Mn}_3\text{O}_4$ . The reference sample,  $\text{Mn}_3\text{O}_4$ , shows the typical expected Raman bands with bands at 287.9, 317.3, 373.4, 657.6  $\text{cm}^{-1}$ , and an additional weak band with a maximum at 570.5  $\text{cm}^{-1}$ . A prominent band at 652.5  $\text{cm}^{-1}$  assigned to  $A_{1g}$  symmetric stretching of the  $\text{M-O}$  bond of the  $\text{MnO}_6$  octahedra unequivocally confirms the formation of  $\text{Mn}_3\text{O}_4$  nanoparticles with a spinel structure.<sup>[12]</sup> The atypical shoulder discernible at 662.3  $\text{cm}^{-1}$  is attributed to structural distortion resulting from intimate interaction of impurity atoms with the spinel structure. The low intensity bands at 229.5, 347.5, and 421.3  $\text{cm}^{-1}$  are characteristic of  $\text{Mn-N}_4$  and arise from stretching or deformation of out-of-plane  $\text{Mn-N}$  Raman modes,<sup>[13]</sup> confirmed by recording the Raman spectrum of

manganese phthalocyanine under similar conditions (Figure 4S). These observations indicate that  $\text{Mn-N}$  bonds are apparently conserved in the final catalyst.

The ability of  $\text{Mn}_x\text{O}_y/\text{NC}$  to electrocatalyze both ORR and OER was investigated by hydrodynamic voltammetry in oxygen-saturated KOH (0.1M) and compared with commercial  $\text{Mn}_2\text{O}_3$ ,  $\text{Mn}_3\text{O}_4$ , and  $\text{MnO}_2$  (Figure 2a), and with  $\text{IrO}_2$ ,  $\text{RuO}_2$ , and  $\text{Pt/C}$  (Figure 2b).



**Figure 2.** Linear-sweep voltammograms showing the electrocatalysis of oxygen reduction and water oxidation by a)  $\text{Mn}_x\text{O}_y/\text{NC}$  and commercial manganese oxides, b)  $\text{Mn}_x\text{O}_y/\text{NC}$  and  $\text{Pt/C}$ ,  $\text{IrO}_2$ , and  $\text{RuO}_2$ , c)  $\text{Co}_x\text{O}_y/\text{NC}$  compared with  $\text{Co}_3\text{O}_4$  and d)  $\text{Ni}_x\text{O}_y/\text{NC}$  compared with  $\text{NiO}$ , in oxygen saturated KOH (0.1 M) at a scan rate of 10  $\text{mV s}^{-1}$  and rotation of 1600 rpm. GC = Glassy carbon.

$\text{Mn}_x\text{O}_y/\text{NC}$  clearly shows a higher oxygen-evolution current at all potentials compared to ordinary manganese oxides. The reduction of oxygen by  $\text{Mn}_x\text{O}_y/\text{NC}$  starts at a much lower overpotential and the reduction current is pronouncedly higher than that of any of the conventional oxides. We attribute the ability of  $\text{Mn}_x\text{O}_y/\text{NC}$  to catalyze both ORR and OER with much higher efficiency than ordinary manganese oxides to its unique structure and composition. The porous nitrogen-rich carbon shell surrounding the  $\text{Mn}_x\text{O}_y$  particles confers enhanced conductivity to the particles thus concertedly promoting their activity for both OER and ORR. Concomitantly, the ORR activity is further augmented by the nitrogen-functionalized carbon groups in NC.

Interestingly, the method also works very well when applied to synthesize corresponding cobalt,  $\text{Co}_x\text{O}_y/\text{NC}$  (Figure 2c) and nickel,  $\text{Ni}_x\text{O}_y/\text{NC}$  (Figure 2d) catalysts, which outperform  $\text{Co}_3\text{O}_4$  and  $\text{NiO}$ , respectively, for both ORR and OER.

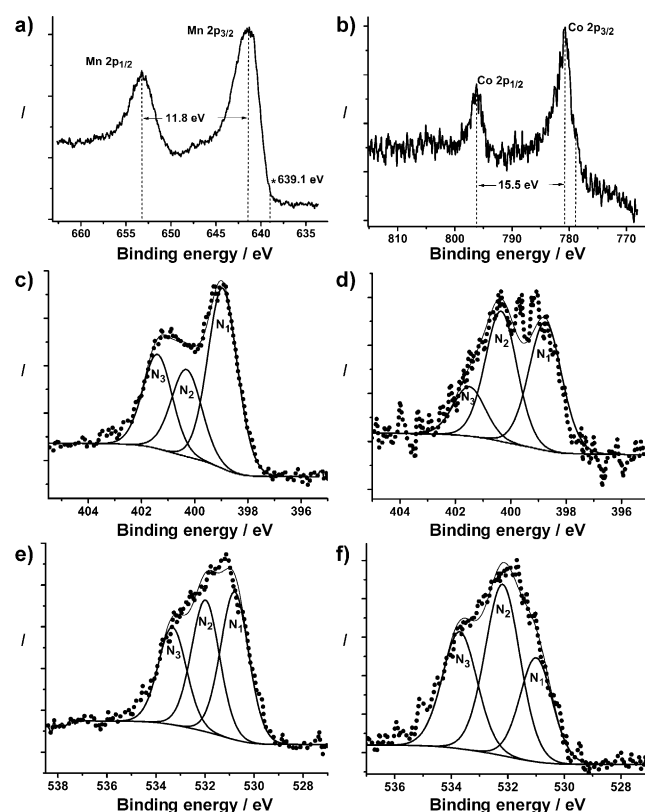
The overvoltage between ORR and OER translates into loss in efficiency and is therefore a very important parameter for evaluating the bifunctional electrocatalytic activity of a given catalyst (Table 1). The OER activity was computed as

**Table 1:** The bifunctional activity of various catalysts for ORR and OER.

Catalyst	$E_{\text{ORR}}/\text{V}$ 1 $\text{mA cm}^{-2}$	$E_{\text{OER}}/\text{V}$ 10 $\text{mA cm}^{-2}$	$\Delta E/\text{V}$ ( $E_{\text{ORR}} - E_{\text{OER}}$ )
$\text{RuO}_2$	0.54	1.64	1.10
$\text{IrO}_2$	0.38	1.70	1.32
$\text{Pt/C}$	0.96	1.90	0.94
$\text{Mn}_x\text{O}_y/\text{NC}$	0.81	1.68	0.87
$\text{Ni}_x\text{O}_y/\text{NC}$	0.71	1.64	0.93
$\text{NiO}$	0.32	1.74	1.42
$\text{Co}_x\text{O}_y/\text{NC}$	0.80	1.66	0.86
$\text{Co}_3\text{O}_4$	0.27	1.68	1.41

the potential at a current density of  $10 \text{ mA cm}^{-2}$ , which has been proposed as a figure-of-merit for the viability of a given catalyst in solar fuel synthesis.<sup>[14]</sup> The ORR activity of the catalysts was compared at the potential corresponding to a current density of  $1 \text{ mA cm}^{-2}$  during oxygen reduction. Strikingly, the low overvoltage between ORR and OER would translate into an energy saving of at least 80 mV, 70 mV, and 10 mV, respectively, if instead of Pt/C, the  $\text{Co}_x\text{O}_y/\text{NC}$ ,  $\text{Mn}_x\text{O}_y/\text{NC}$ , and  $\text{Ni}_x\text{O}_y/\text{NC}$  bifunctional catalysts are used.

The chemical interaction of the various components of  $\text{Mn}_x\text{O}_y/\text{NC}$  and  $\text{Co}_x\text{O}_y/\text{NC}$  was probed by X-ray photoelectron spectroscopy (XPS). The Mn 2p signals (Figure 3a) have doublets at 641.4 eV (Mn 2p<sub>3/2</sub>) and 653.3 eV (Mn 2p<sub>1/2</sub>)



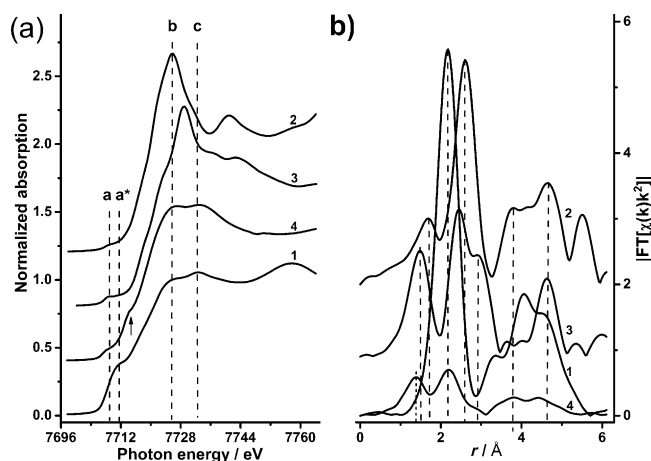
**Figure 3.** XPS spectra showing the 2p core-level spectra of Mn in  $\text{Mn}_x\text{O}_y/\text{NC}$  (a) and Co in  $\text{Co}_x\text{O}_y/\text{NC}$  (b), N 1s spectra of  $\text{Mn}_x\text{O}_y/\text{NC}$  (c) and  $\text{Co}_x\text{O}_y/\text{NC}$  (d), and high-resolution O 1s spectra of oxygen in  $\text{Mn}_x\text{O}_y/\text{NC}$  (e) and in  $\text{Co}_x\text{O}_y/\text{NC}$  (f).

corresponding to a spin-orbit splitting of 11.80 eV typical of  $\text{Mn}_3\text{O}_4$ .<sup>[15]</sup> The weak shoulder at 639.1 eV, untypical of manganese oxides, is ascribed to Mn–N<sub>x</sub> moieties as confirmed by Raman spectroscopy. The Co 2p core-level spectrum (Figure 3b) is devoid of any shake-up peaks and the Co 2p<sub>1/2</sub> and Co 2p<sub>3/2</sub> bands are centered at 780.7 eV and 796.2 eV with a separation of 15.5 eV between them, typical of  $\text{Co}_3\text{O}_4$  in very good agreement with well-characterized  $\text{Co}_3\text{O}_4$ .<sup>[16]</sup> The broad shoulder centered at 778.8 eV is uncharacteristic of cobalt oxides and oxyhydroxides, or of  $\text{Co}^0$  whose signal is expected at  $(778.1 \pm 0.02) \text{ eV}$ .<sup>[16]</sup> The Co–N<sub>4</sub> core has been observed to be conserved during pyrolysis of cobalt-containing N<sub>4</sub>-macrocyclic complexes at low temperatures ( $< 700^\circ\text{C}$ ).<sup>[17,18]</sup> We therefore ascribe this shoulder to the presence of Co–N<sub>x</sub> moieties in  $\text{Co}_x\text{O}_y/\text{NC}$ . The N 1s spectra of both  $\text{Co}_x\text{O}_y/\text{NC}$  and  $\text{Mn}_x\text{O}_y/\text{NC}$  were deconvoluted into three peaks. The co-existence of nitrogen-doped carbon in the catalysts complicates unambiguous assignment of the peaks. However, a literature survey indicates that the unusually sharp intensity of the low-binding-energy signals of the N 1s spectra of both  $\text{Mn}_x\text{O}_y/\text{NC}$  (Figure 3c) and  $\text{Co}_x\text{O}_y/\text{NC}$  (Figure 3d) cannot be exclusively due to the commonly encountered nitrogen-functionalized carbon groups, namely pyridinic, pyrrolic, and quaternary, expected at 398.6, 400.3, and  $(401.3 \pm 0.3) \text{ eV}$ , respectively,<sup>[19,20]</sup> indicating that a fraction of the metal centers apparently remain coordinated to nitrogen.

The O 1s region of  $\text{Mn}_x\text{O}_y/\text{NC}$  was deconvoluted into three different contributions at 530.8, 532.2, and 533.7 eV. The sharp peak at a binding energy of 530.8 eV is due to lattice oxygen ( $\text{O}^{2-}$ ) of  $\text{Mn}_3\text{O}_4$ . The lower intensity peaks at 532.1 eV and 533.4 eV are due to oxygen-containing functional groups on the carbon surface.<sup>[20]</sup> The O 1s region of  $\text{Co}_x\text{O}_y/\text{NC}$  was also deconvoluted into three contributions at 531.0 eV and 532.2 eV assigned to coordinatively unsaturated oxygen species and surface hydroxylation respectively, and at 533.7 eV assigned to oxygen groups on the carbon surface (Figure 3f).<sup>[21]</sup>

The X-ray near edge structure (XANES) of  $\text{Co}_x\text{O}_y/\text{NC}$  shows the spectral features typical of both metallic cobalt and CoO. The pre-edge region, formed by  $1s \rightarrow 3d$  electron transition is represented with two peaks (marked “a” and “a\*” in Figure 4a) at 7709 eV and 7711 eV. The 7709 eV peak is in perfect agreement with the pre-edge peak position of the CoO reference spectrum, while the second peak is shifted by approximately 3 eV relative to XANES of metallic cobalt which is due to Co–N as observed for cobalt phthalocyanine encapsulated in multi-walled carbon nanotubes.<sup>[22]</sup> Above the absorption edge, there are two peaks of relatively low intensity at 7725 and 7732 eV (“b” and “c” in Figure 4a). Both energies coincide with distinctive features of CoO and cobalt foil. The lower energy peak is, however, considerably more intense than the one for Co foil and therefore formed by overlapping patterns of both references used.

In the extended X-ray fine structure (EXAFS, Figure 4b), there are indications of both light backscattered and cobalt being present in the first coordination sphere. The low distance peak at  $r = 1.4 \text{ \AA}$  is presumably formed by both Co–O and Co–N backscattering events. Although the presence of nitrogen in the first coordination sphere is not clearly

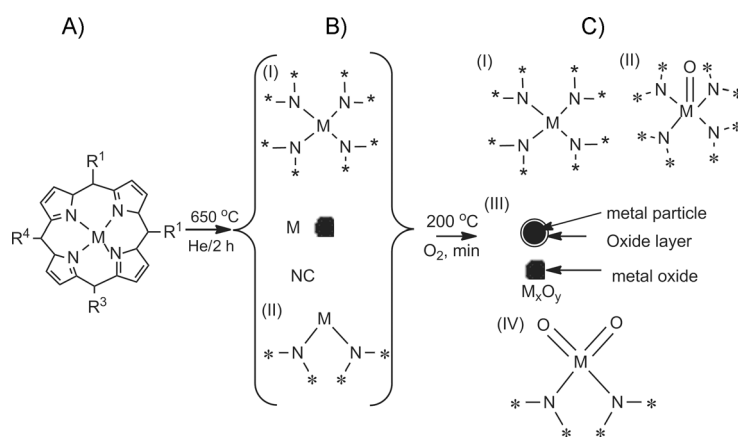


**Figure 4.** a) XANES, and b) EXAFS spectra of  $\text{Co}_x\text{O}_y/\text{NC}$  (4) and references Co foil (1), CoO (2), and  $\text{Co}_3\text{O}_4$  (3). The shoulder at about 7714.1 eV (see arrow) confirms intimate interaction between Co and N and thus the existence of Co–N<sub>x</sub> moieties.

evident, it can be distinguished by the broader and lower peak intensity indicating the interference of two waves with almost equal period, but shifted in phase to form a not perfectly favorable interference pattern (see Figure 8S).

Fitting of the EXAFS spectrum in the *k*-space shows that the local environment of Co can be better described by a three-shell model including Co–O, Co–N, and Co–Co scattering events (goodness of fit  $R=9.6\%$ ) rather than the two-shell model which includes only oxygen and cobalt as first-coordination-shell neighbors (goodness of fit  $R=14.5\%$ ). The corresponding fit parameters are provided in Table 2.

These observations lead us to conclude that during the pyrolysis step, the complexes degrade but the  $\text{MN}_4$  core of the metallomacrocylic ring is largely conserved (Scheme 1), and, the metallic species in B undergo oxidation to form metal oxides or surface oxide species during the calcination step. Structure B(I) is a known active site for ORR and is obtained by pyrolysis of  $\text{N}_4$ -macrocylic complexes  $< 700^\circ\text{C}$ , particularly those containing Fe and Co.<sup>[18,23]</sup> Structures C(II) and C(IV) are variants of B(I) and B(II), respectively, and are expected to be thermodynamically more stable than their B counterparts, since their formation results in saturation of



**Scheme 1.** Schematic illustration of the possible groups that are formed during pyrolysis of cobalt and manganese  $\text{N}_4$ -metallomacrocylic complexes at  $650^\circ\text{C}$  for 2 h under He, and subsequent calcination in oxygen at  $200^\circ\text{C}$  for 30 min.

the coordination capacity of the metals. We suppose that the unique simultaneous and interactive presence of nitrogen functionalized carbon groups, Mn–N<sub>x</sub> and Co–N<sub>x</sub> moieties, and the spinels of the respective metals are responsible for the remarkable capability of  $\text{Mn}_x\text{O}_y/\text{NC}$  and  $\text{Co}_x\text{O}_y/\text{NC}$  to catalyze both ORR and OER with such exceptional efficiency.

In particular, the nitrogen-doped carbon surrounding the metal oxide nanoparticles concertedly enhances the electrocatalytic performance of the oxides by improving their conductivity and complementarily catalyzing the ORR. The long-term performance of the catalysts was evaluated using double-pulse chronopotentiometry, with each pulse lasting one hour and alternating between oxygen evolution and oxygen reduction (Figure 9S). The catalysts last a few cycles then rapidly degrade owing to physical detachment of the catalyst films caused by the gas-bubble effect. Generally,  $\text{Mn}_x\text{O}_y/\text{NC}$  was the least stable, while  $\text{Co}_x\text{O}_y/\text{NC}$  and  $\text{Ni}_x\text{O}_y/\text{NC}$  had similar stability, maintaining a fairly constant over-

voltage between oxygen reduction and oxygen evolution before degradation occurred.

In conclusion we uncovered a method for synthesis of inexpensive catalysts for bifunctional oxygen electrodes with potential application in metal–air batteries, unitized regenerative fuel cells, and alkaline electrolyzers.  $\text{Mn}_x\text{O}_y/\text{NC}$  and  $\text{Co}_x\text{O}_y/\text{NC}$  showed much lower overvoltage between the OER and ORR reactions compared to the state-of-the-art catalysts  $\text{RuO}_2$ ,  $\text{IrO}_2$  and Pt/C (Pt (20%) on carbon), which places them among the best bifunctional catalysts for oxygen electrodes reported to date.

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**Table 2:** CoK-edge EXAFS fitting using 3- and 2-shell models.<sup>[a]</sup>

	<i>r</i> [Å]	CN	<i>R</i> [%]
<b>3-shell model</b>			
Co–O	1.99	2.31	9.6
Co–N	1.88	1.45	
Co–Co	2.52	2.04	
<b>2-shell model</b>			
Co–O	$1.90 \pm 0.02$	$3.57 \pm 0.42$	14.5
Co–Co	$2.54 \pm 0.01$	$1.92 \pm 0.32$	

[a] *r* = distance from the central Co to the backscattering atom, CN = apparent coordination number, *R*-factor, indicating goodness of fit.

**Keywords:** heterogeneous catalysts · nanoparticles · nitrogen-doped carbon · oxygen reduction · water oxidation



- [1] G. Girishkumar, B. McCloskey, A. C. Luntz, S. Swanson, W. Wilcke, *J. Phys. Chem. Lett.* **2010**, *1*, 2193–2203.
- [2] a) K. M. Abraham, *J. Electrochem. Soc.* **1996**, *143*, 1; b) S. W. Lee, C. Carlton, M. Risch, Y. Surendranath, S. Chen, S. Furutsuki, A. Yamada, D. G. Nocera, Y. Shao-Horn, *J. Am. Chem. Soc.* **2012**, *134*, 16959–16962.
- [3] J.-S. Lee, S. T. Kim, R. Cao, N.-S. Choi, M. Liu, K. T. Lee, J. Cho, *Adv. Energy Mater.* **2011**, *1*, 34–50.
- [4] J. Christensen, P. Albertus, R. S. Sanchez-Carrera, T. Lohmann, B. Kozinsky, R. Liedtke, J. Ahmed, A. Kojic, *J. Electrochem. Soc.* **2012**, *159*, R1.
- [5] a) V. Nikolova, P. Iliev, K. Petrov, T. Vitanov, E. Zhecheva, R. Stoyanova, I. Valov, D. Stoychev, *J. Power Sources* **2008**, *185*, 727–733; b) G. Chen, S. R. Bare, T. E. Mallouk, *J. Electrochem. Soc.* **2002**, *149*, A1092; c) Y. Shao, S. Park, J. Xiao, J.-G. Zhang, Y. Wang, J. Liu, *ACS Catal.* **2012**, *2*, 844–857.
- [6] T. Reier, M. Oezaslan, P. Strasser, *ACS Catal.* **2012**, *2*, 1765–1772.
- [7] I. Katsounaros, S. Cherevko, A. R. Zeradjanin, K. J. J. Mayrhofer, *Angew. Chem.* **2014**, *126*, 104–124; *Angew. Chem. Int. Ed.* **2014**, *53*, 102–121.
- [8] L. Jörissen, *J. Power Sources* **2006**, *155*, 23–32.
- [9] F.-D. Kong, S. Zhang, G.-P. Yin, Z.-B. Wang, C.-Y. Du, G.-Y. Chen, N. Zhang, *Int. J. Hydrogen Energy* **2012**, *37*, 59–67.
- [10] F. Cheng, J. Shen, B. Peng, Y. Pan, Z. Tao, J. Chen, *Nat. Chem.* **2011**, *3*, 79–84.
- [11] a) Y. Liang, Y. Li, H. Wang, J. Zhou, J. Wang, T. Regier, H. Dai, *Nat. Mater.* **2011**, *10*, 780–786; b) Z. Jian, P. Liu, F. Li, P. He, X. Guo, M. Chen, H. Zhou, *Angew. Chem.* **2014**, *126*, 452–456; *Angew. Chem. Int. Ed.* **2014**, *53*, 442–446.
- [12] F. Buciuman, F. Patcas, R. Craciun, D. R. T. Zahn, *Phys. Chem. Chem. Phys.* **1999**, *1*, 185–190.
- [13] R. R. Gaughan, D. F. Shriver, L. J. Boucher, *Proc. Natl. Acad. Sci. USA* **1975**, *72*, 433–436.
- [14] Y. Gorlin, T. F. Jaramillo, *J. Am. Chem. Soc.* **2010**, *132*, 13612–13614.
- [15] J. W. Lee, A. S. Hall, J.-D. Kim, T. E. Mallouk, *Chem. Mater.* **2012**, *24*, 1158–1164.
- [16] M. C. Biesinger, B. P. Payne, A. P. Grosvenor, L. W. Lau, A. R. Gerson, R. S. Smart, *Appl. Surf. Sci.* **2011**, *257*, 2717–2730.
- [17] T. Okada, M. Gokita, M. Yuasa, I. Sekine, *J. Electrochem. Soc.* **1998**, *145*, 815–822.
- [18] M. Ladouceur, *J. Electrochem. Soc.* **1993**, *140*, 1974–1981.
- [19] C. V. Rao, C. R. Cabrera, Y. Ishikawa, *J. Phys. Chem. Lett.* **2010**, *1*, 2622–2627.
- [20] S. Kundu, W. Xia, W. Busser, M. Becker, D. A. Schmidt, M. Havenith, M. Muhler, *Phys. Chem. Chem. Phys.* **2010**, *12*, 4351–4359.
- [21] S. C. Petitto, E. M. Marsh, G. A. Carson, M. A. Langell, *J. Mol. Catal. A* **2008**, *281*, 49–58.
- [22] J. C. Swarbrick, T.-C. Weng, K. Schulte, A. N. Khlobystov, P. Glatzel, *Phys. Chem. Chem. Phys.* **2010**, *12*, 9693–9699.
- [23] a) G. Faubert, G. Lalande, R. Cote, D. Guay, J. P. Dodelet, L. T. Weng, P. Bertrand, G. Denes, *Electrochim. Acta* **1996**, *41*, 1689–1701; b) Z. Shi, H. Liu, K. Lee, E. Dy, J. Chlistunoff, M. Blair, P. Zelenay, J. Zhang, Z.-S. Liu, *J. Phys. Chem. C* **2011**, *115*, 16672–16680; c) M. Yuasa, A. Yamaguchi, H. Itsuki, K. Tanaka, M. Yamamoto, K. Oyaizu, *Chem. Mater.* **2005**, *17*, 4278–4281; d) J. Masa, A. Zhao, W. Xia, M. Muhler, W. Schuhmann, *Electrochim. Acta* **2014**, *128*, 271–278.